Full paper

Visualization of facet-dependent pseudo-photocatalytic behavior of TiO₂ nanorods for water splitting using In situ liquid cell TEM

Zu-Wei Yin¹,²,¹¹, Sophia B. Betzler²,¹, Tian Sheng³,⁴, Qiubo Zhang², XinXing Peng²,⁵, Junyi Shangguane, Karen C. Bustillo³, Jun-Tao Li³, Shi-Gang Sun⁴,⁵, Haimei Zheng³,⁶,⁷

¹ College of Energy, Xiamen University, Xiamen, 361005, China
² National Center for Electron Microscopy, Molecular Foundry, Lawrence Berkeley National Laboratory, Berkeley, CA 94720, United States
³ State Key Lab of Physical Chemistry of Solid Surfaces, College of Chemistry and Chemical Engineering, Xiamen University, Xiamen, 361005, China
⁴ College of Chemistry and Materials Science, Anhui Normal University, Wuhu, 241000, China
⁵ National Center for Electron Microscopy, Molecular Foundry, Lawrence Berkeley National Laboratory, Berkeley, CA 94720, United States
⁶ Department of Materials Science and Engineering, University of California, Berkeley, CA, 94720, United States
⁷ Corresponding author. Materials Sciences Division, Lawrence Berkeley National Laboratory, Berkeley, CA, 94720, United States.

Keywords:
In situ TEM
Liquid cell TEM
Photocatalysis
Water splitting
Facet-dependent behavior

ABSTRACT
We report an investigation of the pseudo-photocatalytic behavior of rutile TiO₂ nanorods for water splitting observed with liquid cell transmission electron microscopy (TEM). The electron beam serves as a “light” source to initiate the catalytic reaction and a “water-in-salt” aqueous solution is employed as the electrolyte. The observation reveals that bubbles are generated preferentially residing near the (110) facet of a rutile TiO₂ nanorod under a low electron dose rate (9.3–18.6 e-/Å²·s). These bubbles are ascribed to hydrogen gas generated from the pseudo-photocatalytic water splitting. As the electron beam current density increases to 93 e-/Å²·s, bubbles are also found at the (001) and (111) facets as well as in the bulk liquid solution, demonstrating the dominant effects of water electrolysis by electron beam under higher dose rates. The facet-dependent pseudo-photocatalytic behavior of rutile TiO₂ nanorods is further validated using density functional theory (DFT) calculation. Our work establishes a facile liquid cell TEM setup for the study of pseudo-photocatalytic water splitting and it may also be applied to investigation of other photo-activated phenomena occurring at the solid-liquid interfaces.

1. Introduction
Photocatalytic water splitting has attracted a lot of attention in recent years, since it converts sunlight into clean energy of hydrogen gas. TiO₂ has been widely studied as the active photocatalyst for water splitting, due to its high photocatalytic activity, superior chemical stability and non-toxicity [1]. TiO₂ is a wide-gap semiconductor which exists in many different polymorphs – rutile, anatase being the most common ones. A large variety of anatase and rutile TiO₂ nanostructures have been synthesized for catalytic applications [2]. Since the catalytic reaction only takes place when the photon-induced electrons and holes are available on the surface, significant effort has been made to understand the facet-dependent photocatalytic behavior of TiO₂ nanostructures [3]. First principles calculations indicated that the (101) facet of anatase structure facilitate the reduction reaction while oxidation reactions are preferred on the (001) facet [4]. These were proved by some experimental studies [5,6]. For instance, reductive (101) facets along with oxidative (001) is important for charge separation to enhance photo-degradation of methyl orange [6]. For rutile nanostructures, both (111) and (001) facets seem to promote oxidation reactions, while reductive properties were found for the (110) facets [7,8]. So far, most of the experimental studies were achieved either by monitoring the assemble activity of nanocrystals with different morphologies or through indirect characterizations, such as, the formation of Pt (Pt⁴⁺→Pt⁰; reductive) or PbO₂ (Pb²⁺→Pb⁴⁺; oxidative) nanoparticles on different TiO₂ facets [68].

In situ liquid cell transmission electron microscopy (TEM) has become a powerful tool for the study of liquid-phase reactions and the dynamic phenomena at solid-liquid interfaces [9–11]. It was recently reported that UV light can be introduced into a liquid cell for the study of photo-induced water splitting by TiO₂ nanoparticles. The study showed that a self-hydrogenated shell, generated around the TiO₂ nanoparticle under UV irradiation, can promote photocatalytic H₂ evolution [12]. To limit the electron beam effects, the beam had to be
blanked observe the UV exposure. Thus, it would be a challenge to directly observe the facet-dependent photocatalytic behavior in real time. The work by Roy et al. [13] and Kolmakova et al. [14] demonstrated that inelastic scattering of the electron beam can initiate pseudo-photocatalytic reactions on TiO2 surfaces similar to the ones induced by UV light. Therefore, the electron beam may be used as the “light” source to generate electron-hole pairs, featuring the opportunity to observe pseudo-photocatalytic behavior.

Here, we investigate the facet-dependent pseudo-photocatalytic activity of rutile TiO2 nanorods for water splitting with a facile in situ liquid cell TEM setup. Rutile TiO2 nanorods in a “water-in-salt” solution are encapsulated in a carbon-film liquid cell [35]. Different electron beam dose rates are explored to initiate the water splitting. We find that under a low electron beam dose rate, the facet-dependent pseudo-photocatalytic reactions on the surface of rutile TiO2 nanorods can be achieved. We further conduct theoretical calculation to confirm the experimental observations.

2. Experimental Section

2.1. Synthesis and materials preparation

The raw materials, hydrochloric acid (36.5–38% by weight), titanium butoxide (97%), lithium bis(trifluoromethane sulfonyl) imide (LiN(SO2CF3)2, LiTFSI) (> 99.95%) were purchased from Sigma-Aldrich. Rutile TiO2 was synthesized by a hydrothermal method, as reported elsewhere [26]. LiTFSI was firstly weighed in glove box, then de-ionized water was added to make the 21 M LiTFSI (mol-salt in kg-solution) “water-in-salt” solution [18].

2.2. Liquid cell preparation

Rutile TiO2 nanorods were firstly dispersed in ethanol, then dropped on copper grids with a thin carbon film and dried at 150 °C for 20 min. A droplet (0.5–1 μL) of the as-synthesized solution (21 M LiTFSI or pure water) was subsequently dropped on one TEM grid which was covered by another copper grid with carbon film to make a carbon/liquid/carbon sandwich structure. Then, the liquid cells were left for drying under ambient atmosphere for 1 h. Finally, the samples were transferred into a pre-vacuum system for 2 h to let the carbon film cell seal through van der Waals force.

2.3. In situ TEM

A JEOL 2100 electron microscope operated at 200 kV was used for in situ observation of facet-dependent pseudo-photocatalytic behavior. The dose rates were determined in a separate experiment in the absence of a sample by recreating the exact beam conditions in the in situ experiments. Characterization and statics of TiO2 were performed using FEI Tecnai F20 and Themis 60-300 operated at 200 kV and 300 kV, respectively. The bubbles were marked in the images and assigned to different predefined regions: Region A on top of the nanowire, region B close to the nanowire surface, and region C and D in the electrolyte at different distances from the nanowire surface. The bubble area statics in videos were conducted by Matlab software based on pixel counts. The relative bubble coverage is given as ratio between bubble area in the different regions and the total size of the sub-region.

2.4. Computational methods

All electronic structure calculations were carried out using the Vienna Ab initio Simulation Package with Perdew–Burke–Ernzerhof (PBE) functional of exchange-correlation. The projector-augmented-wave (PAW) pseudopotentials were utilized to describe the core-valence electron interaction with electrons from Ti 3p, 3d, 4s; O 2s, 2p; and H 1s shells [27–30]. The on-site Hubbard U term (U + J) was added on O 2p orbitals at the value of 6.3 eV [31]. The rutile TiO2 (110), (111) and (001) surfaces were modeled as a p(1 × 1) periodic slab with four layers with 8 Ti and 16 O atoms, as shown in Fig. S7. The bottom two layers were fixed and other atoms including adsorbed H2O were relaxed during geometry optimizations. The cutoff energy was set to 500 eV and the occupancy of the one-electron states was calculated by the Gaussian smearing (SIGMA = 0.05 eV). A 6 × 6 × 1 Monkhorst–Pack k-point sampling was used and the Dipole correction was added along the z direction. The adsorption energy was defined in this work as: $E_{\text{ads}} = E(\text{water/surface}) - E(\text{surface}) - E(\text{water})$, where E (water/surface), E(surface), and E(water) are the total energies of the adsorbate binding to surface, free adsorbate in gas phase and clean surface, respectively [32]. The free energy of species was obtained from $G = E + ZPE - TS$, where $E$ is the total energy of species, $S$ is the entropy and $ZPE$ is the zero point energy at room temperature. All the vibrational frequencies, $v_i$ (Hz), were calculated based on the harmonic oscillators approximation [33].

3. Results and discussion

As depicted in Fig. 1a, the carbon-film liquid cell is made with two commercial carbon-coated TEM grids (film thickness 10 nm) encapsulating a liquid thin film. The surface of carbon films are first activated with ozone plasma. Then, a droplet of the “water-in-salt” LiN
(SO$_2$CF$_3$)$_2$ (LiTFSI) aqueous solution or pure water solution containing rutile TiO$_2$ nanorods is dropped onto a carbon film and it is covered with the second carbon film. The subsequent evaporation of water and the van der Waals forces seal a small amount of solution between the two carbon films. A detailed description of the rutile TiO$_2$ nanorod synthesis, the preparation of the aqueous “water-in-salt” LiN(SO$_2$CF$_3$)$_2$ (LiTFSI) solution, and the liquid cell fabrication can be found in the Experimental Section.

The TiO$_2$ nanorods are 70–80 nm in diameter and 500–600 nm long. The ends of a nanorod show sharp tips (Fig. 1b and c). The high resolution images confirm the rutile structure and reveal that a nanorod exposes the (110) facets at its sides and the tips are dominated by the (111) facets, see Fig. 1d as an example. Static analysis confirms that all rutile nanorods expose the (110) facets on the sides and (111) facets at the tip, but not all have the {001} facets at the front of their tips (Fig. S1).

Under a low electron dose rate of 9.3 e-/Å$^2$·s, bubbles can be found on the side of nanorods, but there are no bubbles at the tip facets, even under the prolonged reaction times (70 s) (Fig. 2a, Movie S1). The bubble evolution is analyzed to show the projected area of bubbles as a function of the reaction time. As illustrated in Fig. 2b, the bubble area increases during the initial 30 s of the electron beam exposure. After that the bubble area decreases, which is attributed to the collapse of bubbles (Fig. 2a and c, Movie S1). The observed bubble collapse with time is likely due to an increase of solvated electrons during the prolonged electron beam irradiation (i.e., from the inelastic scattering of the electron beam), which can accumulate on the surface of nanobubbles. The excessive negative surface charges destabilize the nanobubbles [34]. The collapse of nanobubbles has also been observed in previous studies [15,16]. However, as the electron beam dose rate significantly increases (see more details in Fig. 3 as below), the growth of bubbles overcomes the shrinking of the bubbles, which is likely because of the enhanced pseudo-photocatalytic water splitting behavior.

It is noted that slight variations in the behavior of photo-catalytic bubble generation are observed on two sides ((110) facets) of a TiO$_2$ nanorod. This may arise from that the TiO$_2$ nanorod is not perfectly formed adjacent to the (110) side facets of the TiO$_2$ nanorod. However, in contrast to the low dose experiment (9.3 e-/Å$^2$·s) as described before, bubbles are also found in the bulk solution. This effect becomes much stronger at a dose rate of 93 e-/Å$^2$·s, where the bubble concentration in the bulk solution increases significantly and bubbles are found at both the tip and sides of the nanorod. The kinetics of bubble formation is analyzed by subdividing the TEM images into two regions: Region I comprises the TiO$_2$ nanorod and the electrolyte close to its surface, behavior, different dose rates are applied. Characteristic TEM images of the TiO$_2$/aqueous solution system acquired at dose rates of 18.6 e-/Å$^2$·s and 93 e-/Å$^2$·s are displayed in Fig. 3a and c, respectively. The corresponding dynamic bubble generation behavior can be seen in Movie S2, in which the electron dose is increased from 18.6 e-/Å$^2$·s to 93 e-/Å$^2$·s at 40 s. At a dose rate of 18.6 e-/Å$^2$·s, the bubbles are still predominantly formed adjacent to the (110) side facets of the TiO$_2$ nanorod. However, in contrast to the low dose experiment (9.3 e-/Å$^2$·s) as described before, bubbles are also found in the bulk solution. This effect becomes much stronger at a dose rate of 93 e-/Å$^2$·s, where the bubble concentration in the bulk solution increases significantly and bubbles are found at both the tip and sides of the nanorod. The kinetics of bubble formation is analyzed by subdividing the TEM images into two regions: Region I comprises the TiO$_2$ nanorod and the electrolyte close to its surface,
Region II covers the bulk solution in the rest of the image. The relative bubble coverage for each region is evaluated by the ratio of bubble area to the total area. More details of the measurements for Region II (further divided into three sub-regions) are shown in Fig. S3a. The bubble formation kinetics in the two regions is also analyzed under different dose rates (Fig. 3b and d). Under an electron dose rate of 18.6 e-/Å²·s, the relative bubble coverage close to the nanorod in Region I increases during the first 15 s before reaching a steady state with around 20% coverage. The bubble concentration in Region II of the bulk solution is distinctly lower, but shows a continuous increased from 1% to 8% during the course of experiment. When the electron dose rate is increased to 93 e-/Å²·s, the relative bubble coverage in Region I is jumped from 21% to 59% in 15 s. Simultaneously, the bubble coverage in Region II increases to 31%. Additionally, the fine analysis shows that there is no obvious variation within Region II (Fig. S3b). Under all dose rates the relative bubble coverage is significantly higher close to the TiO₂ nanorod than in the bulk solution. We summarize that under a low dose (e.g., 9.3 e-/Å²·s and 18.6 e-/Å²·s), we have only observed bubble generation on the (110) facets but not at the tip facets ((001) and (111)). Under the high dose rate of 93 e-/Å²·s, bubble generation on all facets can be found. The quantification of bubble evolution on different facets under the high dose rate is shown in Table S1.

Supplementary video related to this article can be found at https://doi.org/10.1016/j.nanoen.2019.05.068.

To investigate the role of aqueous “water-in-salt” solution in the facet-dependent bubble generation behavior, a solution of TiO₂ nanorods in pure water is studied as comparison. Fig. 4a shows a typical TEM image of TiO₂ in pure water using the low electron dose of 9.3 e- /Å²·s, which is drastically different from that of TiO₂ in an aqueous “water-in-salt” solution (Fig. 4b). In contrast to what is observed for the “water-in-salt” solution, where bubbles only form on the surface of the TiO₂ nanorods, in pure water bubbles are generated on the TiO₂ surface as well as in the bulk solution (Movie S3 and Movie S1). As depicted in Fig. 3a and b and similar to the analysis described for Movie S2, each frame of Movie S3 and S1 was subdivided into the TiO₂ related Region I and the bulk solution Region II (which is further subdivided into region B, C, and D for Movie S3, Fig. S4a). For TiO₂ nanorods in pure water, the relative bubble coverage initially increases rapidly for both regions before reaching a steady-state (Fig. 4c). A slight decrease of the coverage towards the end of the movie is assigned to the collapse of a fraction of bubbles [15,16]. The relative bubble coverage is lower in the TiO₂ related Region I compared to that in the bulk solution, which arises from that bubbles on top of/below the TiO₂ nanorod are invisible. If we subtract the nanorod area from Region I we find a similar bubble concentration on the surface of the nanorod and in the bulk solution (Fig. S4b). Even under the minimum electron dose required for imaging of a liquid cell in our instrument (3.1 e-/Å²·s) we find that bubbles are generated at similar rates in the bulk solution as well as on the surface of the nanorod (Fig. S5a, Movie S4; electron dose was changed from 3.1 e-/Å²·s to 9.3 e-/Å²·s at 33 s).

Supplementary data related to this article can be found at https://doi.org/10.1016/j.nanoen.2019.05.068.

Multiple studies reported the electrolysis of water by the electron beam during liquid cell TEM experiments resulting in the formation of bubbles due to H₂ or O₂ generation [17]. We find a clearly enhanced stability of the “water-in-salt” Li(H₂O)₂.5-TFSI solution during electron beam irradiation compared to pure water under the same electron dose rate. This agrees well with the study of Suo et al. [16] who showed a strong interaction between the TFSI⁻ ions and the water molecules in the Li(H₂O)₂.5-TFSI structure resulting in an enlarged electrochemical stability window of 3 V compared to H₂O (1.23 V) [18]. Hence, the use of the “water-in-salt” solution is essential for this study to avoid the vigorous reaction of the electron beam with pure water.

Inelastic scattering of the electron beam in solids features a variety of excitations including interband transitions [19], such as transitions from the valence to the conduction band or to higher energy states. Therefore, the electron beam may act as a “light” source and generate electron-hole pairs which subsequently participate in the pseudo-photocatalytic reactions [13,14]. Our findings demonstrate that the pseudo-photocatalytic water splitting reactions on the TiO₂ nanorod surface exceeds the radiolysis of the electrolyte in a range of electron dose rates, when the “water-in-salt” solution instead of pure water is used. From above, we can conclude that rutile TiO₂ nanorods promote the bubble generation and a facet-dependent activity is discovered with bubbles preferentially forming at the (110) facet. We also find that the exposure time required to initiate the catalytic reaction by the electron beam is much shorter (seconds) than that using the UV light in the previous photocatalytic water splitting experiment conducted in a liquid cell (> 10 h) [12].

A schematic illustration of facet-dependent pseudo-photocatalytic behavior on the surface of rutile TiO₂ nanorods is shown in Fig. 5. The (110) facets at the side and the (111) facets at the tip are exposed to the electron beam. Electron-hole pairs are generated inside the semiconductor TiO₂ as a result of inelastic scattering of the electron beam in the material [19]. The created electron-hole pairs subsequently diffuse to the surface of the TiO₂ to participate in the catalytic reaction [13,14]. Due to the much shorter diffusion length of activated holes (~10 nm) compared to the activated electrons (~100 nm) in rutile TiO₂ [20,21], we consider electrons are the main carrier diffusing to the TiO₂ surface.

We consequently focus our theoretic modeling on an understanding of the behavior of electrons at the different surface facets using density

![Fig. 4. Role of the “water-in-salt” aqueous solution in facet-dependent bubble generation behavior of TiO₂ nanorod. (a) A typical TEM image showing the bubble generation behavior of the system of TiO₂ in pure water under an electron dose rate of 9.3 e-/Å²·s. (b) Bubble generation behavior of TiO₂ in a “water-in-salt” solution under the same electron dose rate of 9.3 e-/Å²·s. (c) The amount of bubbles measured by the relative coverage of bubble area as a function of time in region I and region II corresponding to (a) and (b).](image-url)
most holes may be
facets. The water molecules without dissociation are the most stable state on the \( \{110\} \) facet, the dissociation of water molecules or without water dissociation. The water \( \{110\} \) and \( \{111\} \) of TiO\(_2\) nanorods in an aqueous environment, by considering \( \textit{"facets. It was found that water prefers to dissociate on the \( \{111\} \) layer of water molecules on the surface Ti sites to simulate the water under-coordinated Ti sites, which is consistent with previous reports \[23\]. This reduces their hydrophilicity and \( \{001\} \) facets.}

\[
\begin{array}{ccc}
\text{Considering water dissociation} & \text{(110)} & \text{(111)} & \text{(001)} \\
5.15 \text{ eV} & 7.20 \text{ eV} & 5.93 \text{ eV} \\
\text{No dissociation of water} & 5.15 \text{ eV} & 6.05 \text{ eV} & 5.21 \text{ eV}
\end{array}
\]

Table 1

The calculated work function (\( \Phi \)) for the three different facets (i.e., \( \{100\} \), \( \{110\} \) and \( \{111\} \)) of TiO\(_2\) nanorods in an aqueous environment, by considering the dissociation of water molecules or without water dissociation. The water molecules without dissociation are the most stable state on the \( \{110\} \) facet, while the dissociated water is the most stable state on the \( \{111\} \) and \( \{001\} \) facets.

contribution, pseudo-photocatalytic hydrogen evolution on rutile TiO\(_2\) nanorods in a “water-in-salt” solution under low electron dose (e.g., 9.3–18.6 \text{ e}/\text{Å}^2\text{s}).

Fig. 5. A schematic illustration of the proposed mechanism for the facet-dependent pseudo-photocatalytic hydrogen evolution on rutile TiO\(_2\) nanorods in a “water-in-salt” solution under low electron dose.

4. Conclusions

In summary, we have designed a system of rutile TiO\(_2\) nanorods in “water-in-salt” aqueous solution for the investigation of pseudo-photocatalytic water splitting using liquid cell TEM, where the electron beam is used as the “light” source. The higher electrochemical stability of “water-in-salt” electrolyte compared to that of pure water allows to observe bubble generation from the pseudo-photocatalytic water splitting under a low electron dose rate, where the electrolysis of water is suppressed. Electron beam induced electrolysis of water (also featuring bubble generation) is observed at higher electron dose rates.

Real time observation reveals the facet-dependent bubble generation on the \( \{110\} \) facet surface which is ascribed to pseudo-photocatalytic hydrogen evolution. The lower work function for removing electrons on the \( \{110\} \) facets compared to that on the \( \{111\} \) and \( \{001\} \) facets and the poor interaction of water molecules with the \( \{111\} \) and \( \{001\} \) facets are responsible for the observed facet-dependent behavior. The \( \{110\} \) facet exhibits the lowest energy barrier for the removal of electrons suggesting that the water reduction predominantly take place on this facet. The evolution of oxygen is expected on the \( \{111\} \) and \( \{001\} \) facets, but it is impeded. This is likely due to the lack of hole transfer channels due to the low hydrophilicity of respective facets. In combination with the small diffusion length of holes in TiO\(_2\) their recombination with beam electrons is more probable than the oxidation of water.

Our study presents a new method for the pseudo-photocatalytic semiconductor/liquid interfacial behavior using in situ TEM, at which real time imaging can yield profound insights into photocatalytic mechanisms.

Conflicts of interest

There are no conflicts of interest to declare.

Acknowledgements

This work was funded by the U.S. Department of Energy (DOE), Office of Science, Office of Basic Energy Sciences (BES), Materials Sciences and Engineering Division under Contract No. DE-AC02-05CH11231 within the in situ TEM program (KC22ZH) program. Work at the Molecular Foundry (MF) was supported by the Office of Science, Office of Basic Energy Sciences, of the U.S. Department of Energy under Contract No. DE-AC02-05CH11231. Z. W. Yin acknowledges the financial support from China Scholarship Council (201606310151). S. B. Betzler acknowledges financial support from the Alexander von Humboldt Foundation. J. T. Li thanks the support of NSFC (21621091) and Natural Science Foundation of Fujian Province of China (2015J01063). We acknowledge John Turner at MF for his generous help with the SEM measurements of TiO\(_2\) nanorods.

Appendix A. Supplementary data

Supplementary data to this article can be found online at https://doi.org/10.1016/j.nanoen.2019.05.068.

References


[33] A. Gokhale, S. Kandoi, J. Greeley, M. Mavrikakis, J. Dumesic, Molecular-level de-